



Electrical Resistance Change of Ceramic Matrix Composites in Response to Applied Load and Microstructural Damage

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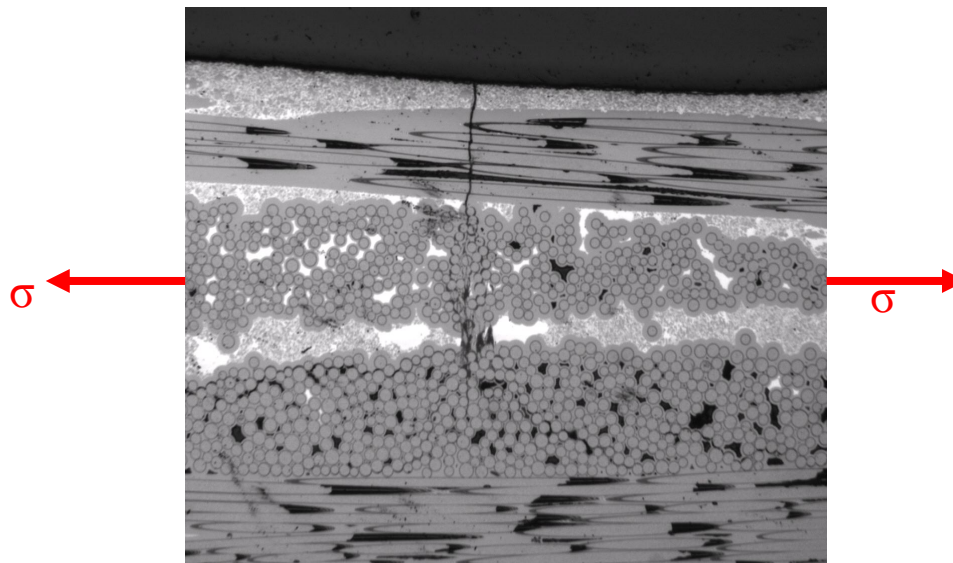
Why Electrical Resistance?

It would be beneficial to accurately detect small-scale transverse matrix cracks in a CMC coupon or component

As cracks form, only bridging fibers can carry current → resistance increases

For MI SiC/SiC, the matrix is more conductive than the fibers, which should give high sensitivity to crack formation

This is a relatively simple technique compared to other inspection methods

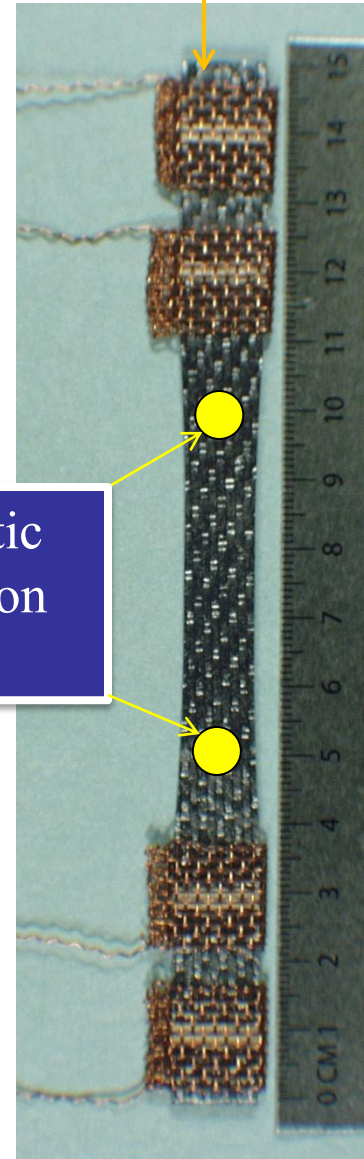




Experimental Procedure

- Resistance measured by four- point probe method
- Silver paint on surface for lower contact resistance
- Grip region wrapped with copper mesh (Used as electrical contact)
 - Offers a simple way of attaching electrical leads for elevated temperature tests
 - For select room-temperature samples, voltage of the contour gage section was also monitored
- Gripped with ceramic wedge grips (for electrical insulation)
- Capacitance strain gage used with 1% range over 25mm
- Resistance monitored with Agilent 34420A micro-Ohm meter
- Acoustic emission monitored by 50kHz to 2MHz sensors just outside the gage section

Copper Mesh



Acoustic
Emission
(AE)



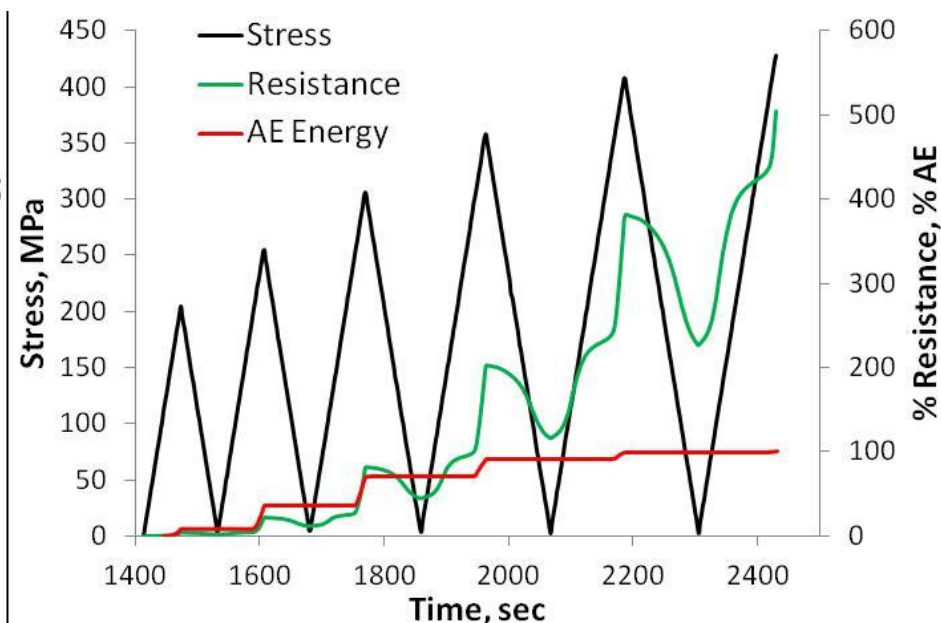
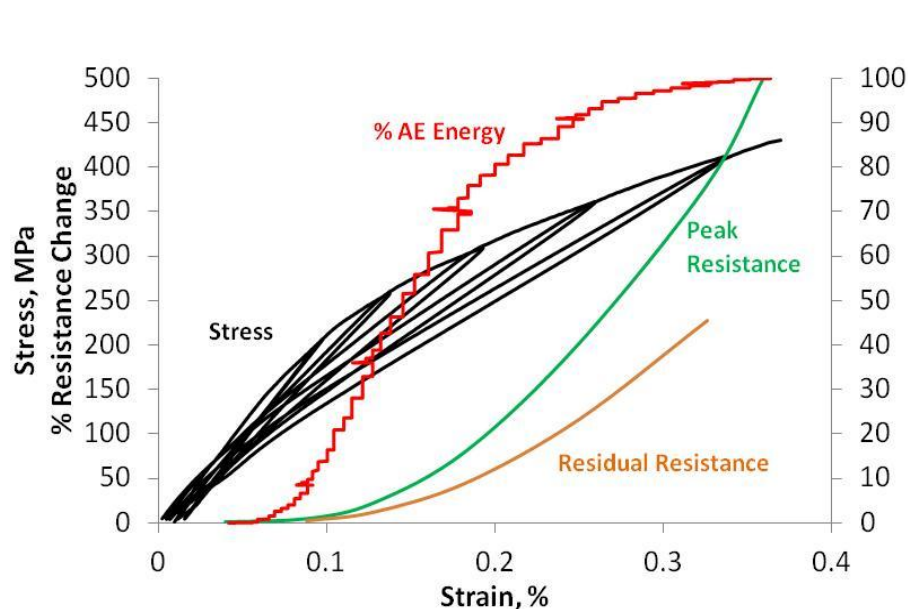
Room Temperature Damage Characterization

Syl-iBN/ Slurry Cast MI Matrix Woven Composite ($f = 0.38$)

Sample	Max Stress, MPa	Stiffness, GPa	Initial Resistivity, $\Omega\text{-cm}$	Resistance change, %	Etched Crack Density, mm^{-1}	PL Stress, MPa	1 st Loud AE Event, MPa	Stress where Resistance is non-linear, MPa
1	247*	246	0.027	17.9	0.9	170	138	140
2	300*	237	0.025	78.5	3	180	132	140
3	430	323	0.026	504	10	NA	103	115
4	440	253	0.023	580	--	180	139	110
5	442	260	0.024	466	11.5	170	147	150

Note: all samples have eight plies with a BN interphase, 800 fibers per tow, and total fiber volume fraction of 0.38

* These samples were unloaded prior to failure to measure crack density

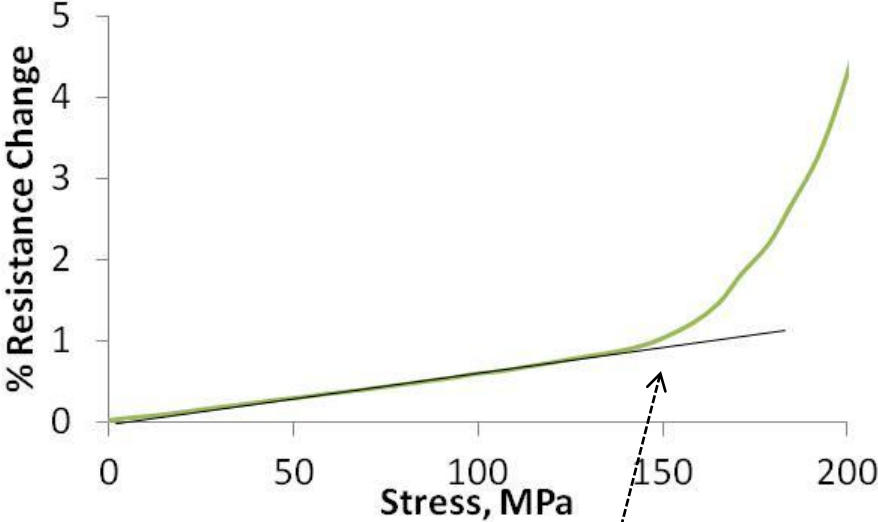




Room Temperature Damage Characterization

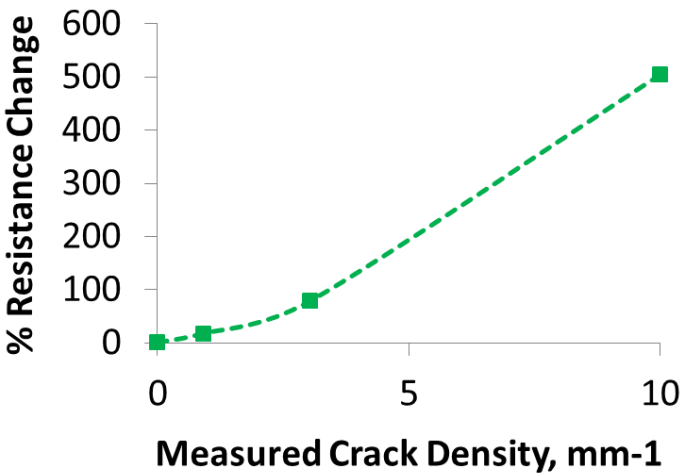
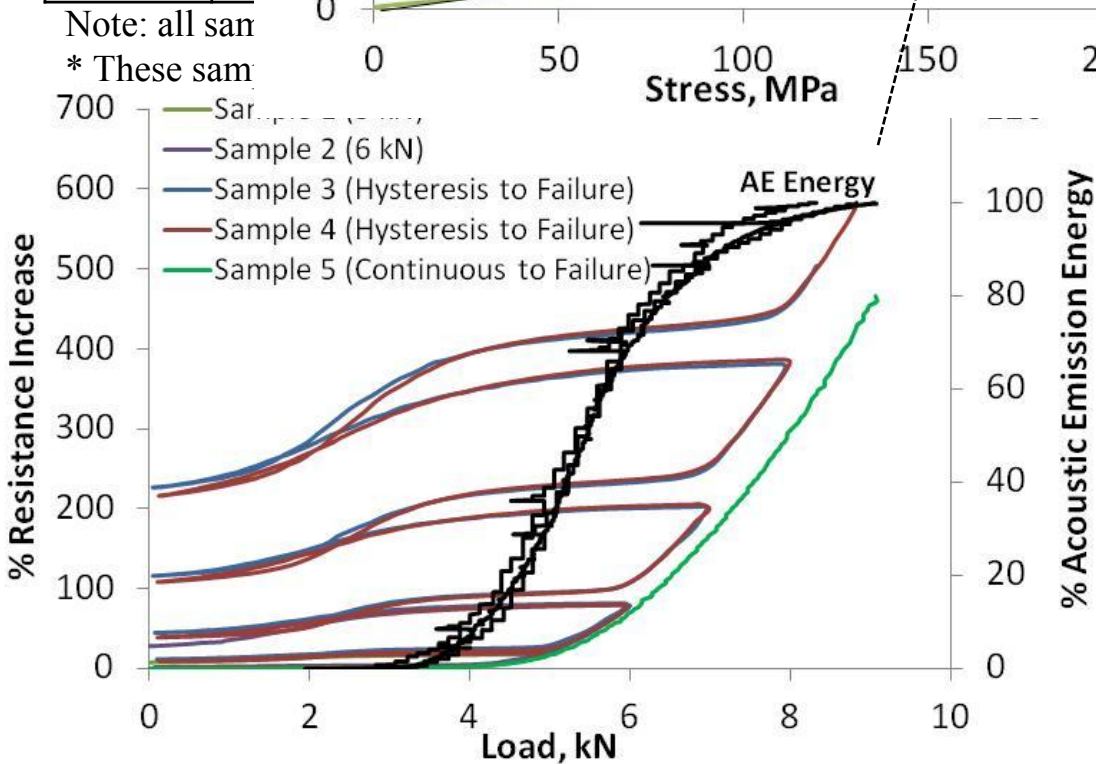
Syl-iBN/ Slurry Cast MI Matrix Woven Composite ($f = 0.38$)

Sample	M St M
1	24
2	30
3	43
4	44
5	44



PL Stress, MPa	1 st Loud AE Event, MPa	Stress where Resistance is non-linear, MPa
170	138	140
180	132	140
NA	103	115
180	139	110
170	147	150

w, and total fiber volume fraction of 0.38



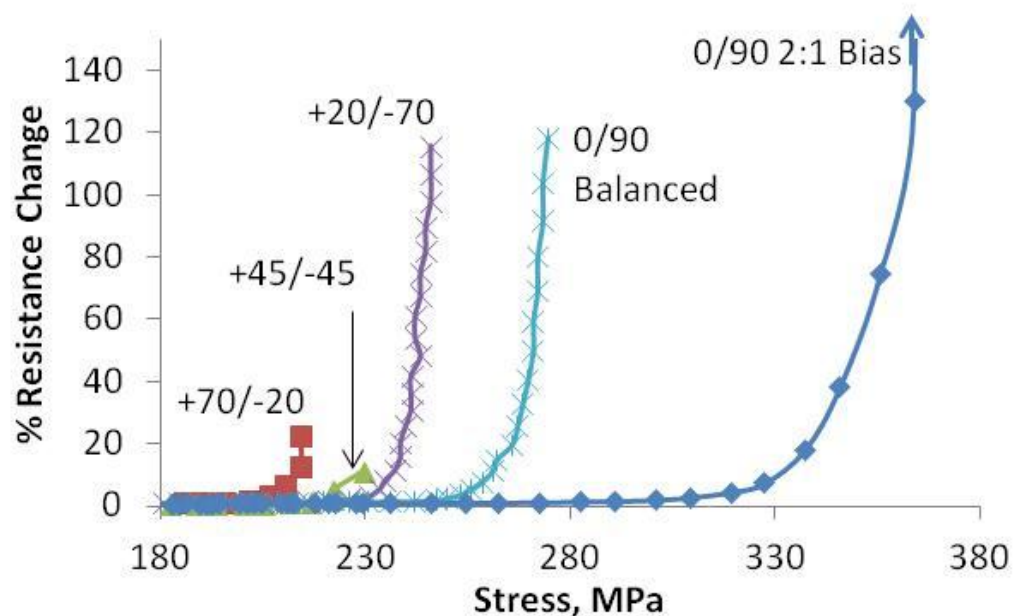
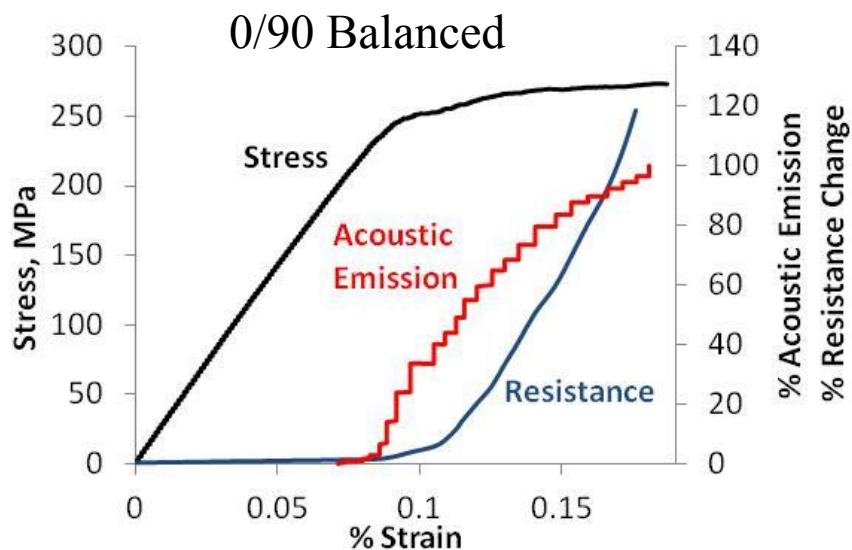


Room Temperature Damage Characterization

GE Hi-NicS/ Pre-preg MI Matrix Laminate composite ($f = 0.22$)

Sample	Max Stress, MPa	Stiffness, GPa	Initial Resistivity, Ω -cm	Resistance change, %	PL Stress, MPa	1 st Loud AE Event, MPa	Stress where Resistance is non-linear, MPa
0/90 balanced (0°)	274	288	0.024	118	248	208	230
0/90 balanced (+/-45°)	228	284	0.027	11.2	225	217	215
0/90 balanced (+20/-70°)	235	282	0.026	129	232	204	210
0/90 balanced (+70/-20°)	210	247	0.027	22.4	210	168	160
0/90 biased 2:1 in 0°	300	292	0.027	530	310	262	260

Note: all samples have eight plies with a BN interphase and total fiber volume fraction of 0.22



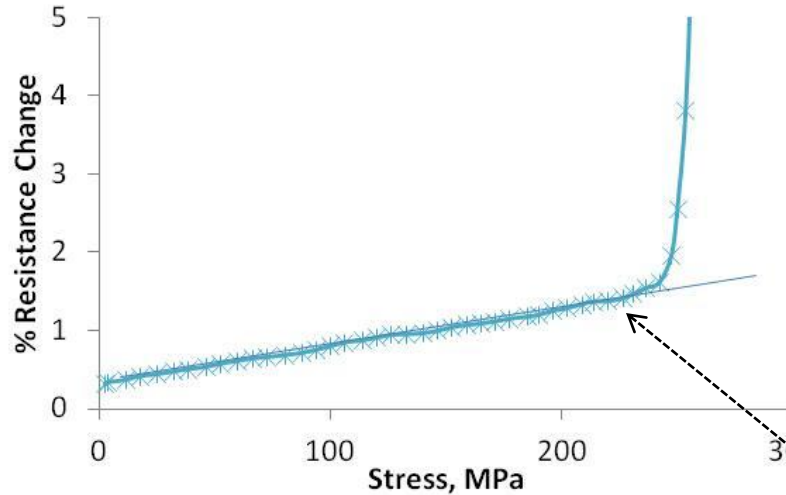


Room Temperature Damage Characterization

GE Hi-NicS/ Pre-preg MI Matrix Laminate composite ($f = 0.22$)

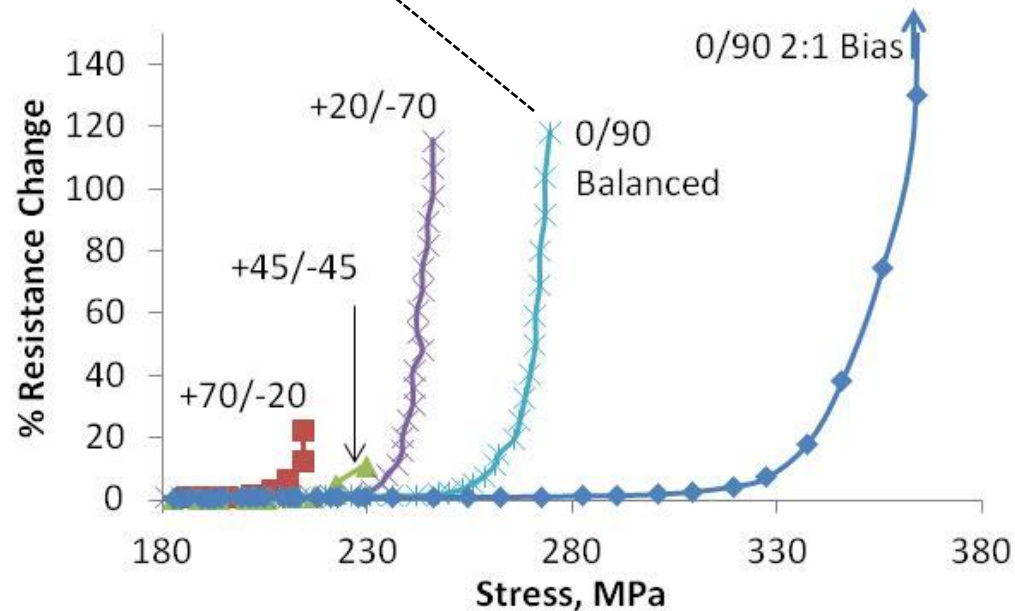
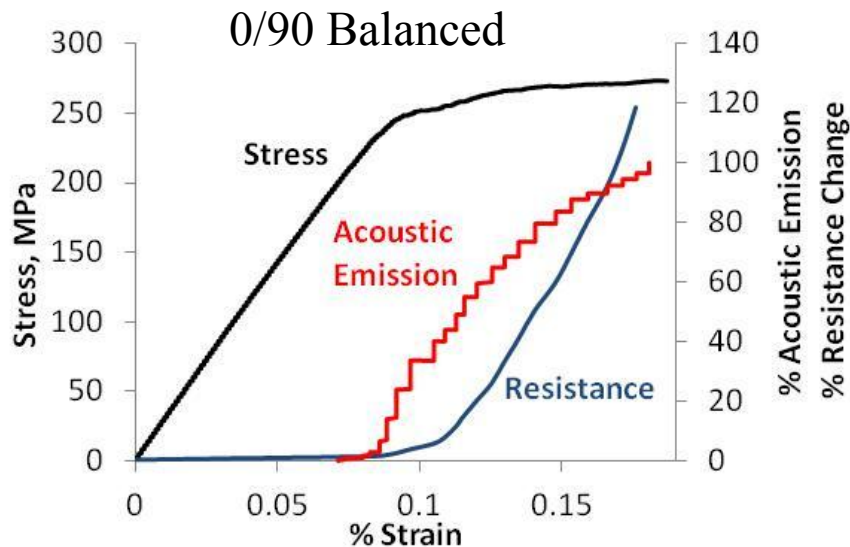
Sample
0/90 balanced (I)
0/90 balanced (II)
0/90 balanced (III)
0/90 balanced (IV)
0/90 biased 2:1

Note: all sample



PL Stress, MPa	1 st Loud AE Event, MPa	Stress where Resistance is non-linear, MPa
248	208	230
225	217	215
232	204	210
210	168	160
310	262	260

Volume fraction of 0.22





The Need for a Model

Key Benefits

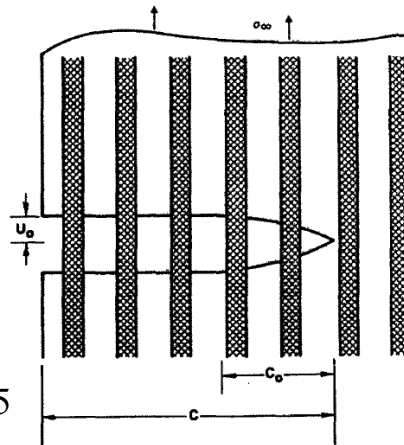
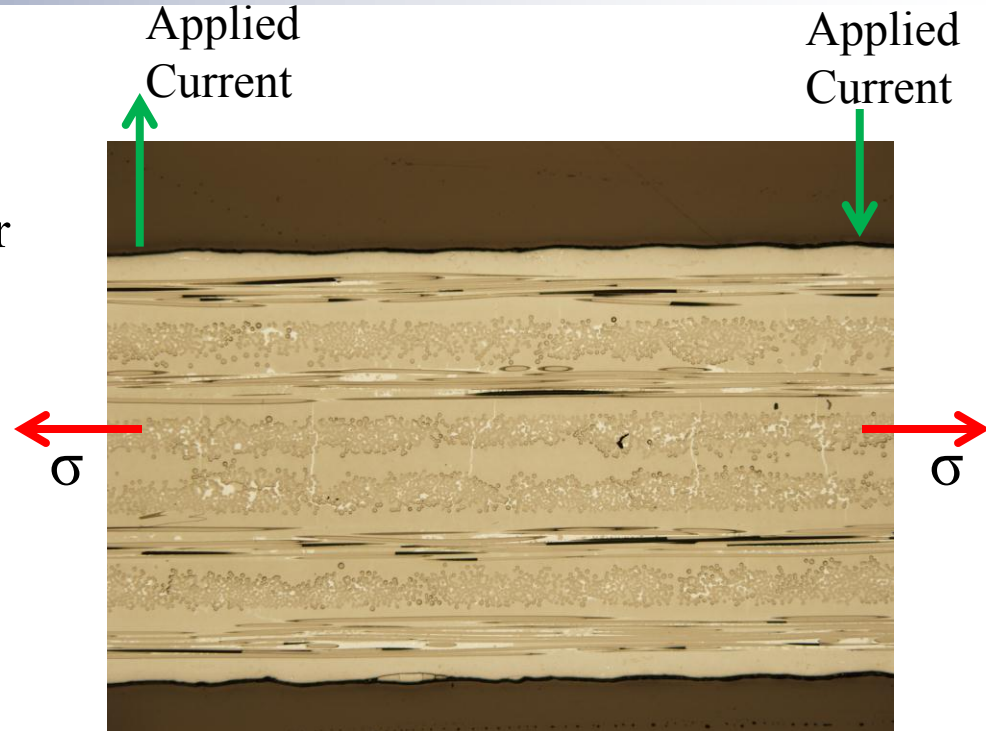
- Can be measured in-situ
- Resistance changes permanently for inspection at zero load
- The deviation from linearity correlates with the proportional limit stress
- Resistance is sensitive to crack formation
- Can be used at elevated temperature
- Repeatable

Key Concerns

- The data needs to be related to the microstructural changes in the material
- Many variables at room temperature
 - crack formation, crack growth, fiber/interphase sliding, fiber breaks
- Even more variables at elevated temp
 - creep of constituents, oxidation, change in resistivity with temperature
- Need to understand what effect each mechanism has on resistance

Model Development

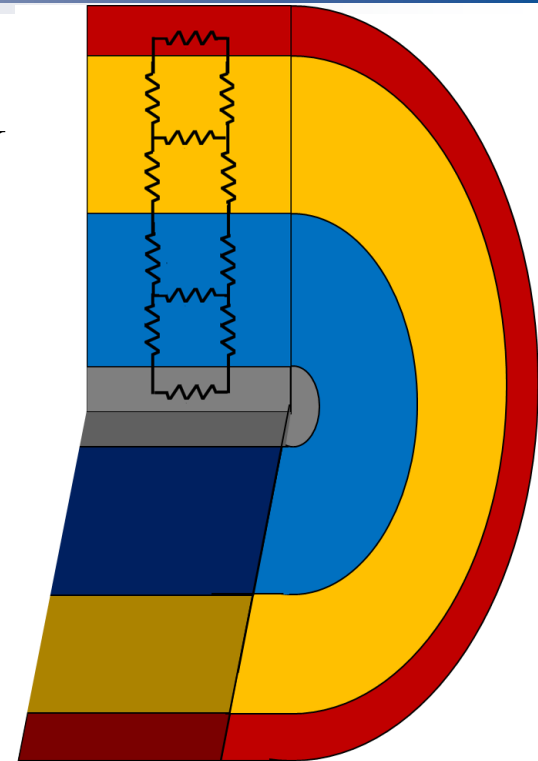
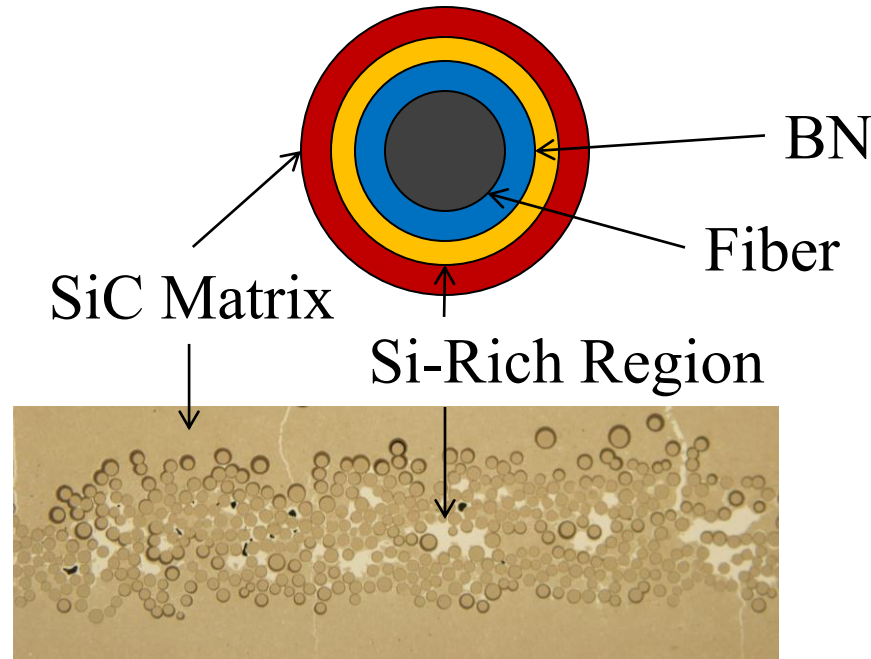
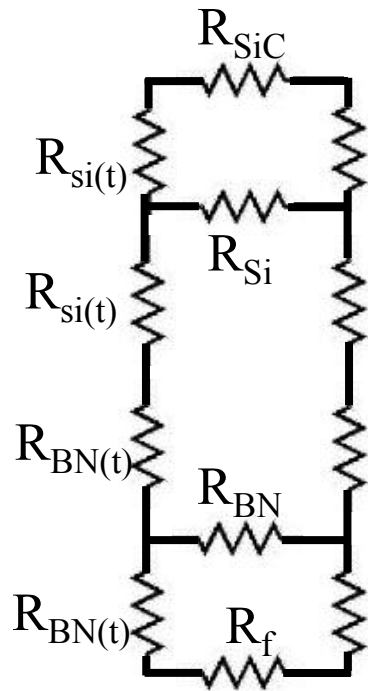
- Current is applied at the MI matrix surface
- The matrix conductivity is an order of magnitude greater than the fibers (MI contains Si)
- Also, BN insulates the fibers from the matrix
- When the matrix is cracked, current must be transferred to the fibers
- The question is how?



Marshall, Cox, Evans 1985



Unit Cell for GE pre-preg MI



- The composite is treated as a series of concentric cylinders
- Fibers are surrounded by BN, then a Si rich region in the ply, then the bulk MI
- All the fibers are treated as one, while maintaining the same relative volume fractions for constituents
- 90° plies are neglected in this case, since the fibers will not bridge the cracks and they do not provide a continuous path (BN interphases in series)
- Each unit cell represents a specified length along the loading direction (10 μm)



Unit Cell for GE pre-preg MI

- Resistance along the length of each constituent:

$$R_c = \frac{\rho A_c}{L}$$

ρ is the constituent resistivity

A_c is the cross-sectional area

L is the length of the unit cell

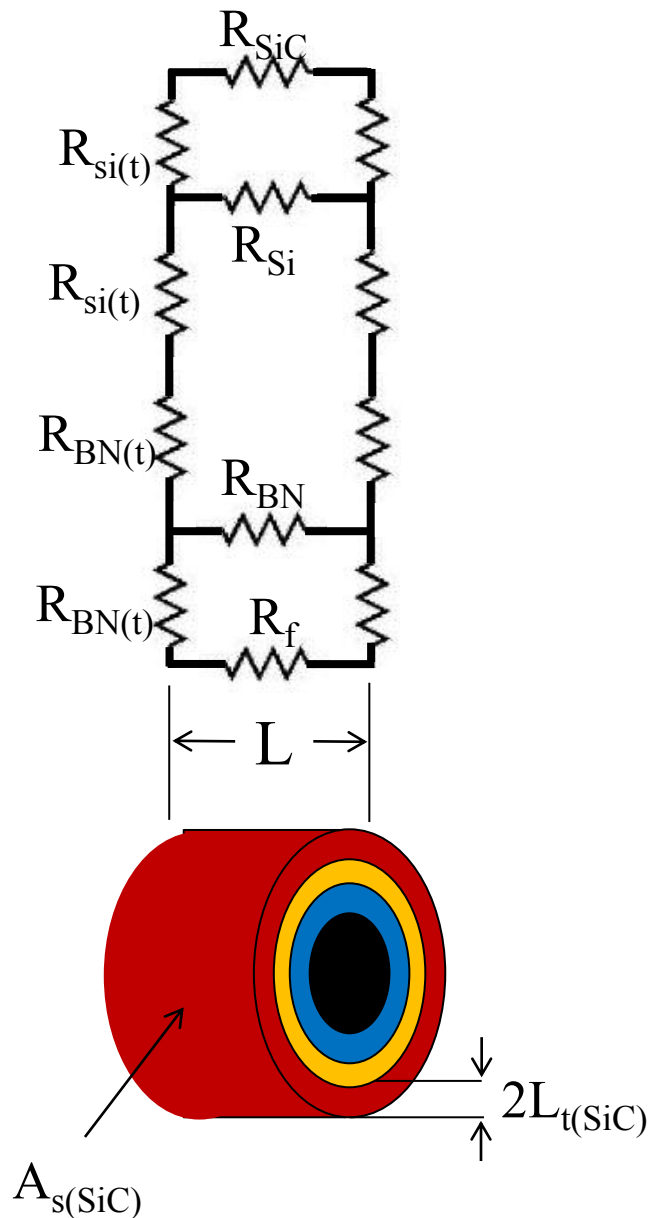
- Similar equations describe the resistance of the constituents in the transverse (radial) direction:

$$R_t = \frac{\rho A_s}{L}$$

A_s is $\frac{1}{2}$ of the mean surface area of the cylinder

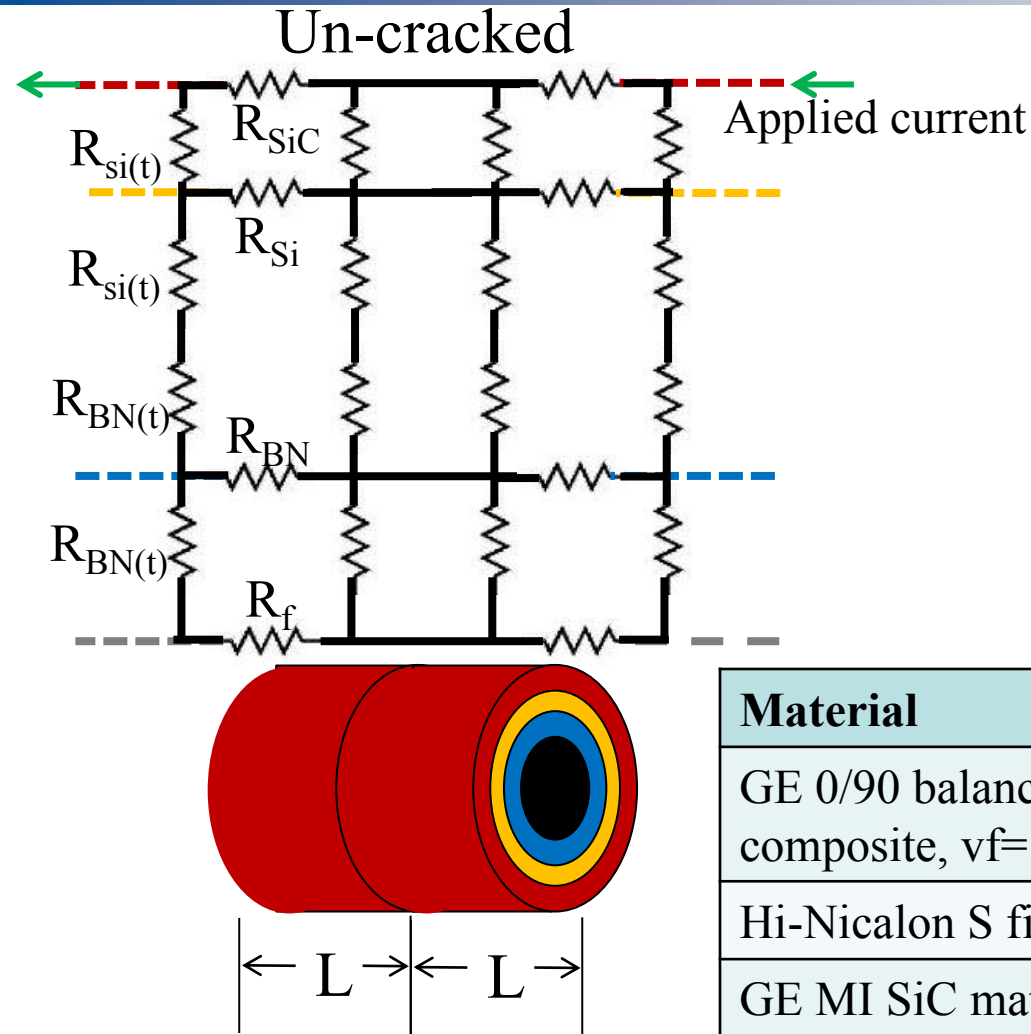
L_t is $\frac{1}{2}$ of the thickness of the cylinder

Material	Resistivity, $\Omega\text{-cm}$
GE 0/90 balanced composite, $v_f = 0.22$	0.027 (measured directly)
Hi-Nicalon S fiber	0.1 (from literature)
GE MI SiC matrix	0.037 (measured directly)
Si rich region in ply	Initially Unknown- very low
BN interphase	Initially Unknown- very high





Model Parameters



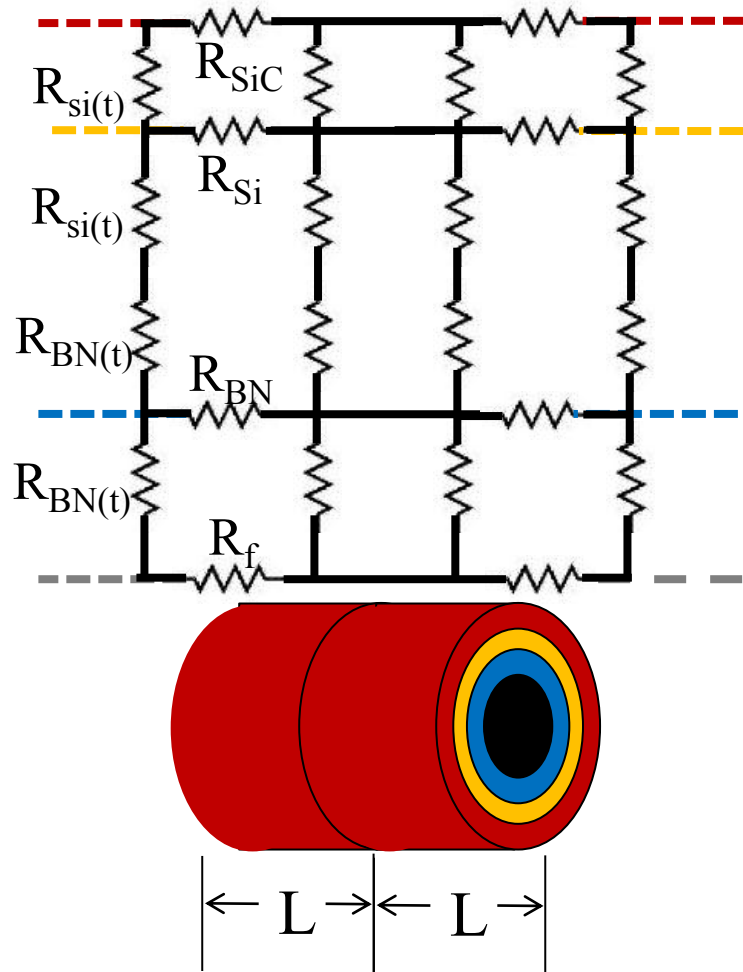
- Current is applied to the first and last cells at the outer SiC layer
- Adjusting the BN resistivity has no effect on the un-cracked composite since the matrix resistivity is low
- Resistivity of the Si-rich region can be adjusted to fit the initial composite resistivity

Material	Resistivity, $\Omega\text{-cm}$
GE 0/90 balanced composite, $v_f = 0.22$	0.027 (measured directly)
Hi-Nicalon S fiber	0.1 (from literature)
GE MI SiC matrix	0.037 (measured directly)
Si rich region in ply	0.0038 (from model)
BN interphase	Initially Unknown- very high

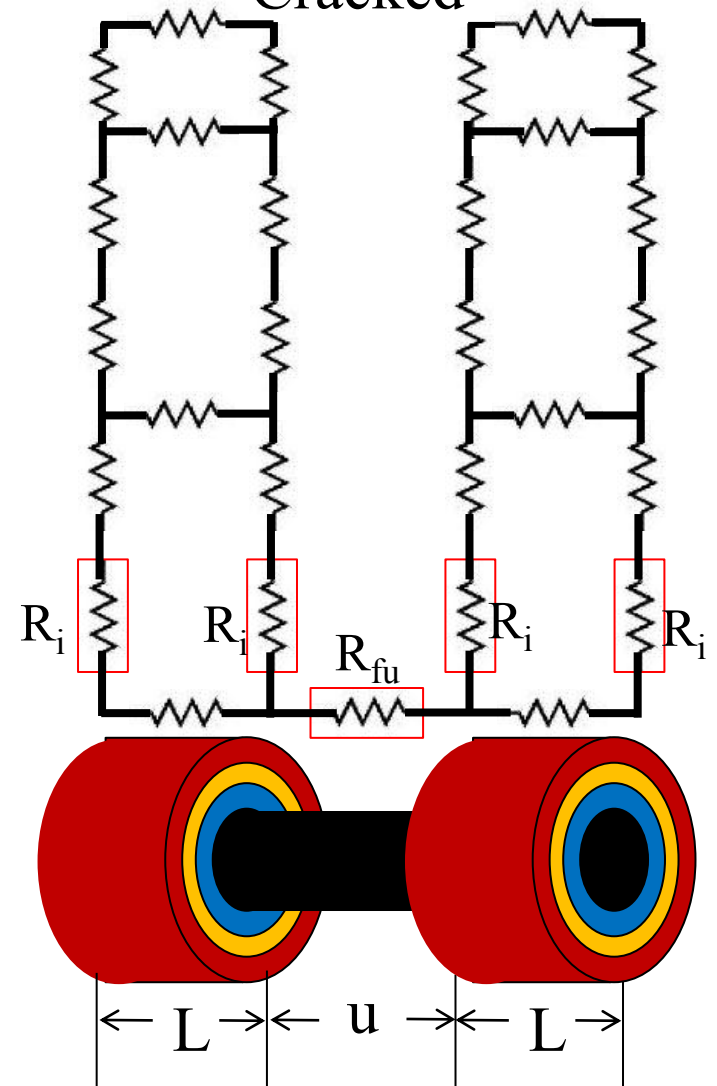


Model Description

Un-cracked



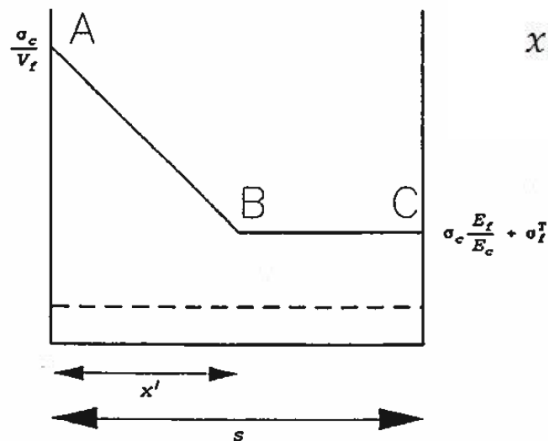
Cracked





Model Description

- The matrix and BN circuits are broken in the crack
- The fiber will de-bond from the BN and slide over a distance x^1
- An interfacial resistance is introduced between the BN and fiber
- The surrounding unit cells within the distance x^1 will all be affected



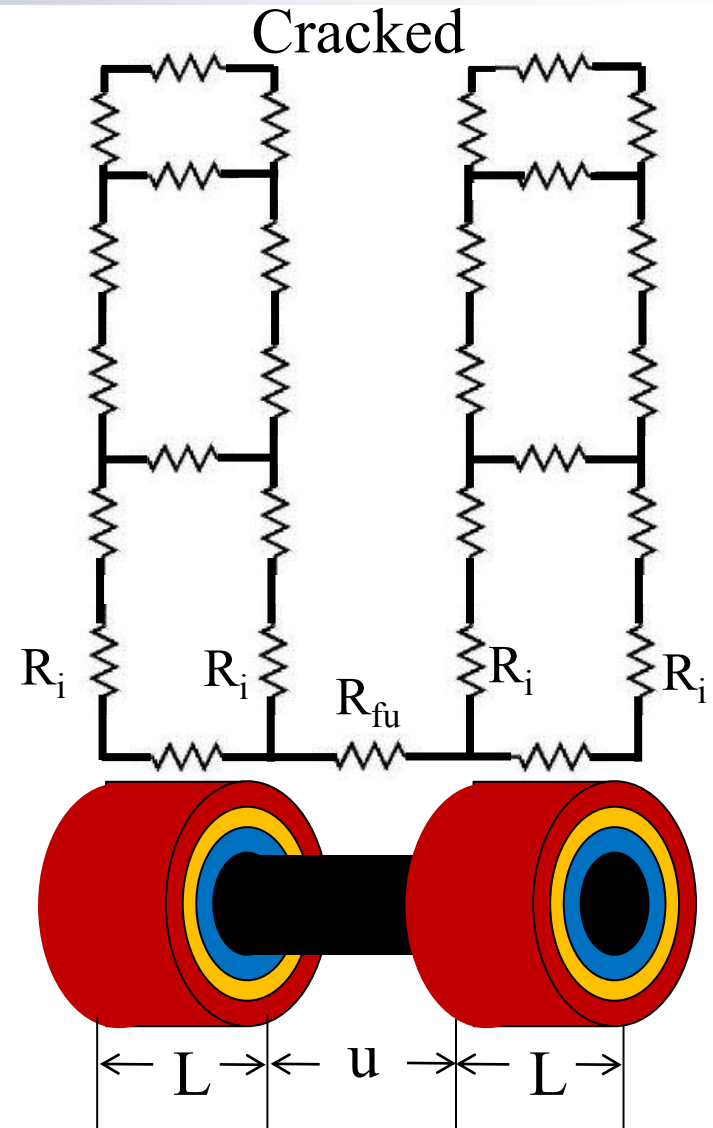
$$x^1 = \frac{r}{2\tau} \left[\sigma_c \frac{v_m E_m}{v_f E_c} - \sigma_f^R \right]$$

Pryce, Smith 1993

8 Stress profile in the fibres for a cracked laminate at an applied stress σ_c

$$u = \frac{\sigma^2 R}{4\tau v_f^2 E_f \left(1 + \frac{E_f v_f}{E_m (1 - v_f)} \right)}$$

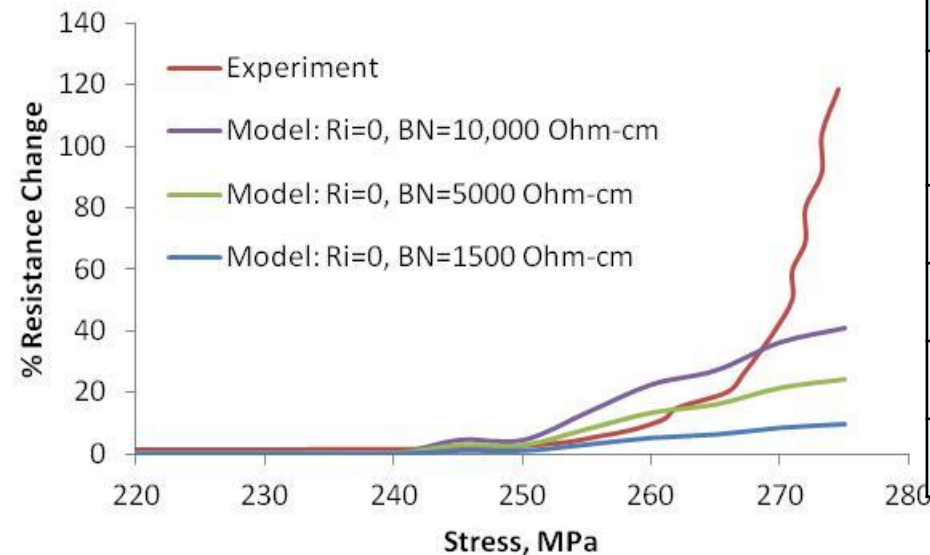
Marshall, Cox, Evans 1985





Model Calibration

- Acoustic emission energy is used as an estimate of crack density as a function of stress
- The model introduces cracks at incremental stress levels, according to the AE data
- The cracks are randomly distributed and the overlap length can be specified
- The resistivity of BN is determined by setting $R_i=0$ and adjusting the BN
- If the value is too high, the model will be too sensitive to cracks
- The maximum BN resistivity was chosen such that the model would never overshoot the experiment in the extreme case of $R_i=0$

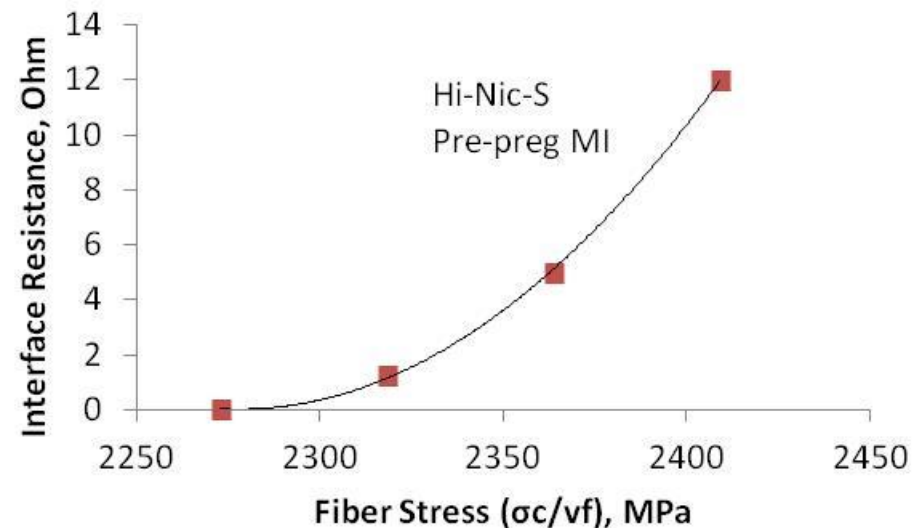
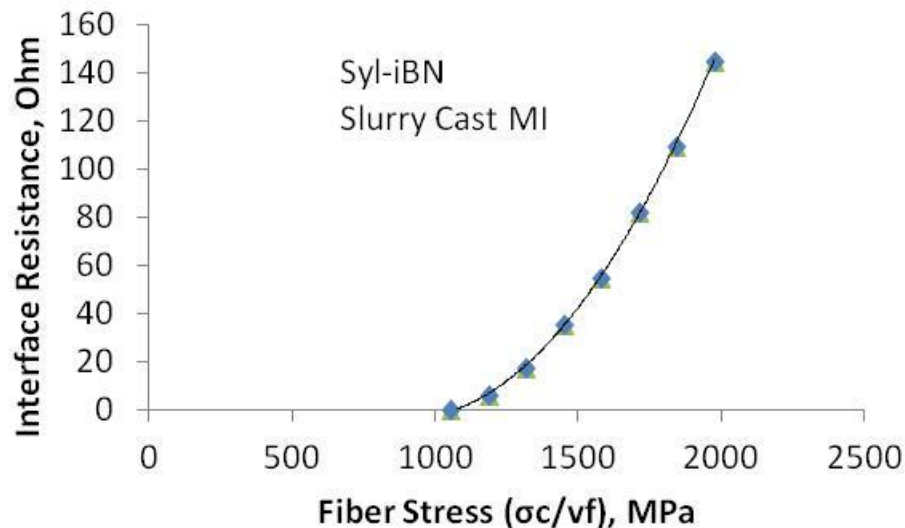


Material	Resistivity, Ω -cm
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Hi-Nicalon S fiber	0.1 (from literature)
GE MI SiC matrix	0.037 (measured directly)
Si rich region in ply	0.0038 (from model)
BN interphase	1500 (from model)



Model Results

- Interfacial resistance R_i is assumed to be uniform over the slip length x^l , but changing with stress
- It is expected that the interfacial resistance increases due to relative sliding between fiber and matrix
- We see that R_i must be proportional to σ^2 to fit the experimental data

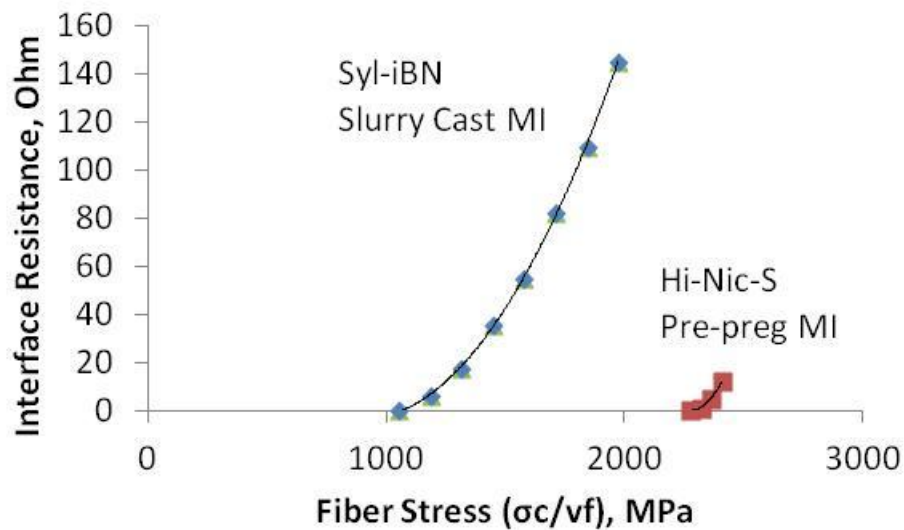


- Sliding distance of the fiber at the crack surface is proportional to σ^2

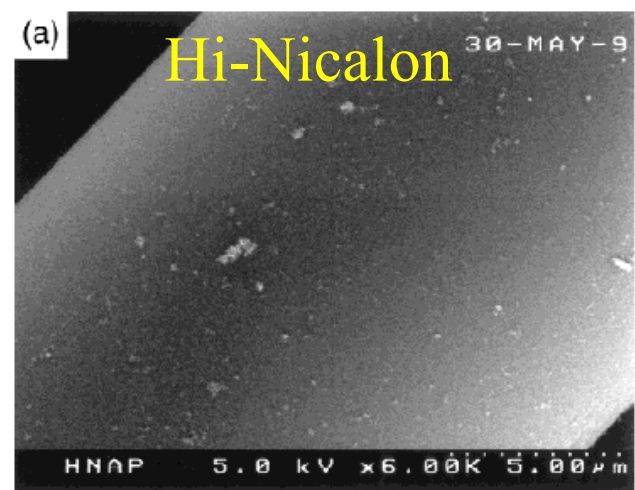
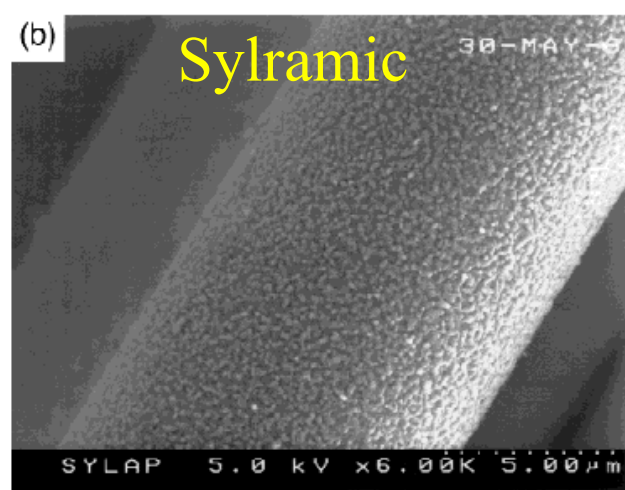
$$u = \frac{\sigma^2 R}{4\tau v_f^2 E_f \left(1 + \frac{E_f v_f}{E_m (1 - v_f)} \right)}$$

Marshall, Cox, Evans 1985

Model Results



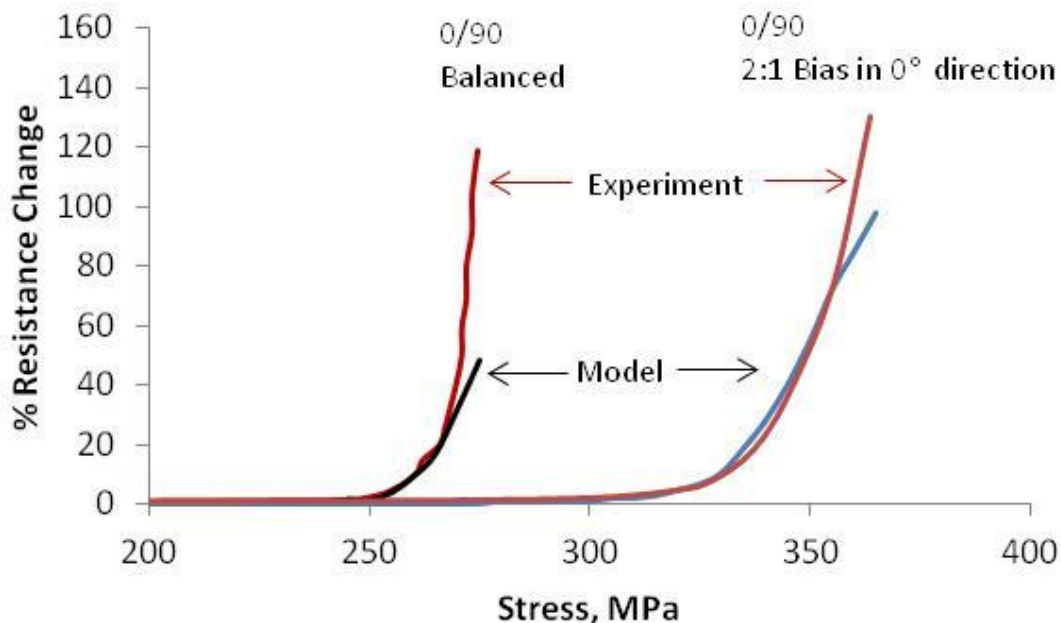
- The difference in magnitude of R_i for the two composites is likely due to fiber roughness





Model Results

- By using the parabolic relationship for R_i , the model fits the experiment for the balanced 0/90 GE pre-preg composite at 98% of the strength
- The same parabolic function generated from the balanced composite was used to model an unbalanced 0/90 sample with 2:1 bias in the loading direction
- The model also fit this data, which indicates that the relationship is more than a mere curve fit





Conclusion

- Electrical resistance offers a way of monitoring damage in CMC's
- Several factors influence the electrical properties
- A discrete model has been developed to understand the mechanisms causing electromechanical changes
- The model verifies that the interfacial resistance is a function of stress squared, consistent with fiber sliding



Future Work

Examine the effect of the following:

- R_i varying along the slip length
- Fiber breaks
- Load-unload- reload
- Varying cross-section along the sample
- Temperature
- Creep
- Environment